Synthesis and Characterization of Robust SiO₂-Phase Change Materials (PCM) Microcapsules

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Abstract

Inorganic SiO₂ microencapsulated phase change materials (MEPCMs) have been developed via polymerization reaction and electrostatic interaction in an emulsion system. The resultant microcapsules were systematically characterized in terms of morphology, composition, thermal stability, durability, and mechanical property. The energy storage capacity of the obtained microcapsules ranges from 190.5 J/g to 225.5 J/g (pure octadecane: 250.6 J/g) when the core material is octadecane at different sizes. SiO₂-PCM microcapsules showed improved energy storage efficiency in terms of shorter melting and freezing durations attributed to the higher thermal conductivity of inorganic silica shells compared with that of polymer shell encapsulated PCM. Repeatable melting and freezing processes after 150 cycles revealed excellent shell tightness and thermal stability of the resultant microcapsules. The high apparent compressive strength of individual SiO₂-PCM microcapsules indicated good survivability in further materials processing for energy storage. The robust SiO₂-PCM microcapsules obtained from a facile fabrication approach in this study have great potential applications for developing energy-efficient materials.

Keywords: Silica shell; Phase change material microcapsules; Energy storage capacity; Energy storage efficiency; Robustness.

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1. Introduction

With the increasing energy demand, there is an urgent need to identify renewable energy sources and invent novel materials for energy storage.^[1,2] Latent heat storage of phase change material (PCM) achieved via a solid-liquid phase transition is a promising technology. With large latent heat storage capacity and a narrow operating temperature range, PCMs provide an efficient way to store thermal energy. PCMs have been widely used in smart textiles to regulate body temperature, in heat

transfer media to save energy, and in solar energy devices for energy storage.^[3,4] Organic PCMs have attracted great attention due to their excellent phase change performance. Among them, N-alkanes (*i.e.* paraffin waxes) are the most promising organic PCMs because they are chemically inert with low vapor pressure, ecologically harmless, noncorrosive, cheap, and widely available. Owing to these desirable characteristics, N-alkanes have been used in various applications.^[1,5-7] Octadecane is an alkane hydrocarbon with a phase change temperature of around 26–28 °C and a high latent heat capacity of 241.2 J/g.^[8] Since the phase change temperature of octadecane is in the thermal comfort range of the human body, octadecane has been used in garments and building products.^[5,6,9-12]

However, direct incorporation of bulk PCM into a matrix such as textile or gypsum wallboard is difficult due to the leaking of PCM during solid-liquid phase change. Packaging of bulk PCM is therefore often necessary so that the melting and solidification process is accomplished in a container that prevents leaking of PCM. In addition, packaged PCM should be compatible mechanically and chemically with the surrounding matrix, possess a stable structure during phase change, and provides a sufficient surface area to encourage heat transfer. Microencapsulation of PCM is there an arrestive of way for PCM packaging. Many studies reported

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Fig. 1 Schematic process of microencapsulation of PCM via deposition of hydrolyzed SiO₂ precursor.

encapsulation of PCM with an organic shell such as polyurea, polyurethane, and polyurea formaldehyde.^[13-15] Wang and Zhang^[16] synthesized the microencapsulated octadecane PCM with a polyurea shell via interfacial polymerization. Li *et al.*^[17] prepared a kind of formaldehyde shell containing PCMs. The organic shell, however, has some major disadvantages including high flammability, poor thermal and chemical stability, and low thermal conductivity. To overcome these shortcomings, PCM encapsulated with an inorganic silica shell was synthesized in an oil and water (O/W) emulsion system^[11-14] via a sol-gel method^[18,19], was currently reported. However, the durability of reported microcapsules is discontented in most of the current studies^[9-11], *i.e.* thermal stability of microcapsules changes significantly after multiple cooling-heating cycles treated. In addition, an essential property, the compressive strength of individual silica-PCM microcapsules, has not been investigated in any articles till now. It is known that the high strength of microcapsules can protect core PCM from external force to ensure no variation of latent heat storage capacity, which makes capsules more practical for real applications.

In this study, a new facile approach to synthesizing SiO₂-PCM microcapsules was developed to address the reported issues as shown in Fig. 1, such as poor thermal stability and weak mechanical strength. Four innovative methods are proposed in this study. First, it is based on the principle of electrostatic adsorption that Si(OH)₄ monomer is adsorbed on a polyurea (PUA) membrane, and then it is further polymerized to produce an inorganic silica shell; second, the outer surface of PCM capsule is a micro-nano hierarchical structure with a larger specific surface area, which is more conducive to the heat exchange of the core PCM; third, the compressive strength test of a single PCM microcapsule is proposed for the first time; finally, the thermal stability of the PCM microcapsule is excellent, the heat enthalpy is basically unchanged after 150 heating and cooling cycles.

In conclusion, a special double-layer structure with an inner polyurea membrane and outer silica shell was innovated and optimized after a parametric study.^[20] The resultant SiO₂-PCM microcapsules were systematically characterized and analyzed in terms of heat storage capacity and efficiency, thermal stability, and mechanical strength.

2. Materials and experiments

2.1 Materials and synthesis of SiO₂-PCM microcapsules

Chemicals including Arabic gum, hexamethylene diisocyanate (HDI), polyethyleneimine (PEI, Mw~1300), hexane, octadecane, tri-methoxymethyl silane (MTTS), and hydrochloric acid solution (HCl, 0.1N) were purchased from Sigma Aldrich and used directly without further purification.

Fig. 2 Schematic diagram of the synthesis of silica-octadecane microcapsules.

As shown in Fig. 2, at ambient temperature, 30 ml of deionized water and 0.93 g of 3 wt.% aqueous solutions of Arabic gum were mixed in a 200 ml beaker. The beaker was suspended in a temperature-controlled water bath on a programmable hot plate with an external temperature probe.

The solution was agitated at 600-1000 rpm with a digital mixer (Caframo) driving a three-bladed propeller. The solution was heated to 50 °C at a heating rate of 5 °C/min. 5 g octadecane and 0.5-1.5 g HDI were then mixed together and slowly added into the aqueous solution to generate emulsion for 10-30 min. After that, 10 g of 1 wt.% aqueous solution of PEI was added dropwise into the emulsion system. After 2 hours of continuous agitation, both the stirrer and hot plate were switched off. The resultant PUA-PCM pre-capsules (octadecane core with a thin layer of polyurea membrane) were washed with distilled water three times. After that, the PUA-PCM pre-capsules were re-dispersed into a beaker loaded with 30 ml of deionized water. MTTs was added into 4 ml aqueous solution (pH=3, prepared using 0.1N HCl) and agitated to initiate hydrolysis at 35 °C for 1 hr. Subsequently, the pre-hydrolyzed MTTs was slowly added into the solution containing PUA-PCM pre-capsules and the solution was agitated at 150 rpm for 1-3 h at a certain temperature (25, 35, or 45 °C) to allow complete deposition of silica on the surface of PUA-PCM pre-capsules through electrostatic interaction. The resultant SiO₂-PCM microcapsules were washed with deionized water and collected after air-drying at room temperature in the fume hood for 24 hours before further analysis.

2.2 Characterizations

Morphology and shell thickness of microcapsules were observed using a field-emission scanning electron microscope (FE-SEM, JEOL, JSM-7600F) equipped with energy-dispersive X-ray spectroscopy (EDS). EDS was used to probe the chemical composition of the shell material. To prepare the sample, a small number of microcapsules was distributed uniformly on a conductive adhesive tape. A few capsules were ruptured by a razor blade in order to observe the core-shell structure. The sample was sputter coated with gold for 40-50 seconds in a vacuum environment and taken out for further analysis.

Fourier transform infrared spectroscopy (FTIR, Varian 3100) was engaged to distinguish the constituents of the SiO₂-PCM microcapsules, octadecane (core material), and the shell. To obtain the shell material, capsule samples were crushed thoroughly and washed three times with hexane to remove the core material (octadecane). After that, the residue (*i.e.* shell material) was filtered and dried in a fume hood at room temperature for 3 hours.

Thermal gravimetric analysis (TGA, Q2950, TA Instruments) and differential scanning calorimetry (DSC, Q200, TA Instruments) were carried out to evaluate the thermal properties of the capsules. TGA on the SiO₂-PCM capsules, octadecane, and the shell was conducted in a nitrogen atmosphere to evaluate the weight change of the sample as a function of increasing temperature up to 600 °C with a constant heating rate of 10 °C/min. An isothermal TGA test on the SiO₂-PCM capsules and octadecane was carried out in a nitrogen atmosphere at 80 °C for 1 hour to evaluate the

thermal stability of the sample as a function of time. DSC on the SiO₂-PCM capsules, the PUA-PCM pre-capsules, and octadecane were evaluated in an N₂ atmosphere with a heating/cooling rate of 10 °C/min. The encapsulation ratio (*R*) and thermal storage capacity (*φ*) of microcapsules can be determined from the following equations.^[21,22]

$$R (\%) = \frac{\Delta H_{m,capsule}}{\Delta H_{m,PCM}} \times 100 \quad (1)$$

$$\phi (\%) = \frac{(\Delta H_{m,capsule} + \Delta H_{c,capsule})/R}{\Delta H_{m,PCM} + \Delta H_{c,PCM}} \times 100 \quad (2)$$

where $\Delta H_{m,capsule}$ and $\Delta H_{c,capsule}$ in J/g are the melting and solidification enthalpy of capsules, respectively. Similarly $\Delta H_{m,PCM}$ and $\Delta H_{c,PCM}$ in J/g are the melting and solidification enthalpy of octadecane, respectively. Enthalpy was calculated by integrating the area under the DSC curve using TA Universal Analysis software. Furthermore, to evaluate the stability and reliability, SiO₂-PCM capsules were undergone 150 heating and cooling cycles in DSC.

The core fraction was obtained by comparing the weight change before and after removing the core material from the microcapsules. A few grams of SiO₂-PCM capsules were collected and weighed first, $W_{capsule}$ in g. The capsule samples were crushed thoroughly and washed three times with hexane to remove the core material (octadecane). After that, the residue (*i.e.* shell material) was filtered and dried in a fume hood at room temperature for 3 hours. The weight of the shell material, W_{shell} in g, was measured and recorded. Core fraction (%) can be calculated with the following Equation. 3. The Core fraction of each sample was measured 3 times and the average and standard deviation were reported.

$$Core\ fraction (\%) = \frac{W_{capsule} - W_{shell}}{W_{capsule}} \times 100 \quad (3)$$

The mechanical strength of the individual capsule with liquid core material was tested under compression with a program-controlled stepper actuator (Physik Instrument M-230S) at a loading rate of 2 μm/s.^[23] A 0.5 N load cell (FUTEK) was used to record the compressive load. Since octadecane with a phase change temperature of 28-30 °C is in a solid state at room temperature (22 °C), microcapsules were heated first to 100 °C for 10 min to ensure the core material remained liquid during the compression test. The apparent mechanical strength of the individual capsule under compression, σ_c in MPa, was estimated using Eqn. 4:^[24]

$$\sigma_c = \frac{P_{max}}{\pi \left(\left(\frac{D_o}{2} \right)^2 - \left(\frac{D_i}{2} \right)^2 \right)} = \frac{4P_{max}}{\pi(2D_o - t)t} \quad (4)$$

where P_{max} in N is the peak load, D_o and D_i in μm are the outer and inner diameter of the capsule, respectively, and t in μm is the shell thickness.

3. Results and discussion

3.1 Mechanism of formation of silica-PCM capsules

In this study, the resultant silica PCM capsules were prepared based on an oil-in-water emulsion system. As can be seen in Fig. 1. Liquid PCM together with HDI as the oil phase is introduced into an aqueous solution containing surfactant. Under agitation, stable oil droplets are formed due to function

Fig. 3 Diameter and the core fraction of SiO₂-PCM capsules as functions of (a) agitation rate; and core fraction and thermal stability of SiO₂-PCM capsules as functions of (b) HDI mass, (c) emulsion time, (d) reaction temperature, and (e) reaction time.

Table 1. DSC thermal properties of octadecane and SiO₂-PCM capsules with different sizes.

Sample	Melting process		Solidification process		Encapsulation ratio (%)	Thermal storage capacity (%)
	T _m (°C)	ΔH _m (J g ⁻¹)	T _c (°C)	ΔH _c (J g ⁻¹)		
Octadecane	27.1	250.6	27.0	227.9	-	-
SiO ₂ -PCM capsule@100μm	27.3	210.2	26.9	188.1	83.9	99.2
SiO ₂ -PCM capsules@300μm	27.4	223.5	27.1	202.3	89.2	99.8
SiO ₂ -PCM capsules@500μm	27.5	230.3	27.1	208.1	91.9	99.7

Note: T_m and T_c are melting and solidification temperatures, respectively. ΔH_m and ΔH_c are melting and solidification enthalpies, respectively.

of surfactant. PEI aqueous solution is then added drop wise. Temperature is set as 50 °C to initiate the reaction between HDI and PEI for 2 hours. During this process, a positively charged polyurea (PUA) membrane is formed around the droplets (*i.e.*, PUA pre-capsules). It is known that PEI molecules contain a long chain of NH₂ group which is positively charged in an aqueous solution. Therefore, negatively charged pre-hydrolysis of MTTs (Si(OH)₄ monomer) tends to be attracted to the PUA membrane by means of electrostatic interaction.^[25,26] Adjusting the pH of the above solution to 3.0 and extending the reaction for a certain time at ambient temperature, a dense and solid silica shell begins to form and grow. Finally, a well-defined and complete silica shell is obtained.

Fundamentally, the formation of silica shell relies on the deposition of MTTs precursor on the surface of PUA membrane via electrostatic force because negatively charged monomer Si(OH)₄ has the tendency to be attracted by and deposits on the positively charged polyurea surface which further forms Si-O-Si (*i.e.* silica) network structure. A fine balance between hydrolysis and condensation of MTTs precursor is necessary to ensure a well-defined core-shell structure of the SiO₂-PCM microcapsules. The properties of resultant capsules were significantly affected by the synthesis conditions such as agitation rate. Specifically, proper pre-hydrolysis of MTTs and pH value play significant roles in the formation of the silica shell. A suitable acidic condition is beneficial for the hydrolysis of MTTs, resulting in accelerating the hydrolysis rate of MTTs. Meanwhile, the condensation rate of hydrolyzed monomer is inhibited.^[21]

3.2 Parametric study

Five key factors, including agitation speed (600-1000 rpm), HDI dosage (0.5-1.5 g), emulsion time (10-30 mins), reaction temperature (25-45 °C), and reaction time (1 hour-3 hour), were systematically studied to reveal their influence on core fraction and thermal stability of the resultant SiO₂-PCM microcapsules. Reference synthesis conditions are an agitation speed of 900 rpm, HDI dosage of 1 g, emulsion time of 10 min,

reaction temperature of 35 °C, and reaction time of 3 hours. Only one factor is changed every time and the other factors remain as the reference synthesis conditions.

Figure 3a shows the diameter and core fraction of SiO₂-PCM capsules under different agitation rates. As can be seen, the size of the microcapsules decreases from 500 μm to 100 μm and the core fraction reduces from 89.5 wt.% to 75.9 wt.% when the agitation speed increases from 600 rpm to 1000 rpm, respectively. It reveals that the core content of capsules was increased with the capsule size enlarged. Probably, it is ascribed to the lower specific surface area of larger capsules, resulting in less quantity of Si(OH)₄ monomer deposited on the capsule surface.

In addition, the encapsulation ratio can be obtained based on the enthalpy of SiO₂-PCM capsules.^[27] In which enthalpy calculated from DSC, it is known that silica is not a phase change material. Therefore, the total enthalpy of SiO₂-PCM capsules is attributed to core PCM. Table 1 gives the enthalpy of samples and their corresponding encapsulation ratio. It shows that the encapsulation ratio increased from 83.9 wt.% to 91.9 wt.% when the capsule diameter increased from 100 μm to 500 μm. In addition, a physical experiment was carried out to validate the results obtained through the enthalpy method. As shown in Fig. 3a, it shows the core fraction increased from 75.9 wt.% to 89.5 wt.%, when the diameter of the capsules increased from 100 μm, 300 μm, and 500 μm. Comparing these two results obtained from two different approaches indicates that the calculation of the enthalpy approach shows good agreement with experimental results.

It is well known that latent heat can only be stored by core PCM instead of the silica shell. Hence, the content of core PCM is the only factor that dominates the encapsulation ratio. Thermal storage capacity stands for the service efficiency of PCM during the phase change process. Normally, encapsulation ratio and thermal storage capacity are used to characterize the phase change performance, and phase change enthalpy has a heavy relationship with the encapsulation ratio (R) and thermal storage capacity (φ) of silica-PCM capsules. Based on equation (1), the theoretical encapsulation ratio and

thermal storage capacity can be obtained as listed in Table 1. In which the highest encapsulation ratio of silica-PCM capsules is about 91.9 wt.% at a diameter of 500 μm , this value is approximately equivalent to the data 89.5 wt.% at the same size investigated by practical experiment as shown in Fig. 3a. These two comparable approaches demonstrate that of silica-PCM capsules have a high encapsulation ratio at a proper capsule size obtained by regulating agitation rate, indicating that high latent heat storage capacity of silica-PCM capsules can be obtained successfully. Similarly, thermal storage capacity can be calculated using equation (2), all of the thermal storage capacities of capsules are higher than 99%, meaning that almost all of the encapsulated core PCM is functional and active for thermal storage.

Figure 3b reveals that the core fraction of SiO_2 -PCM microcapsule reduces significantly from 88.1 to 45.5 wt.% as the HDI amount increases from 0.5 to 1.5 g. This phenomenon is because HDI mixed with PCM as the core materials, PCM mass is a constant 5 g, when HDI mass increased from 0.5 g to 1.5 g, the percentage of PCM (core materials: PCM&HDI) decreased gradually. Capsules synthesized with 1 g HDI exhibit the best thermal stability with only 3.4 % loss in weight, followed by capsules synthesized with 1.5 g HDI (6.9 % weight loss) and with 0.5 g HDI (9.2 % weight loss), after 1 h isothermal heating at 80 $^{\circ}\text{C}$. The optimal HDI dosage of 1 g in the current study allows the formation of positively charged polyurea membrane capsules, enhances the deposition of nano-silica particles, and forms a dense silica shell. At the lower dosage of HDI, the formation of positively charged polyurea membrane capsules is hindered and deposition of nano-silica particles is limited, resulting in loose shell structure and low thermal stability. When higher HDI is used, surrounding water molecules can diffuse into the capsule, react with excessive HDI to form a polyurea inner shell, and simultaneously generate carbon dioxide which can cause porous and loose shell structure, resulting in the reduction of thermal stability of capsules.

As shown in Fig. 3c, the core fraction of capsules remains largely the same at different emulsion times with a core fraction around 62 wt.%. Furthermore, the thermal stability of capsules decreases with increasing emulsion duration as 3.4 wt.% weight loss was observed for the capsules subject to 10 min emulsion while 8 wt.% weight loss was registered for the capsules subject to 30 min emulsion. This may be ascribed to the consumption of HDI due to reaction with water molecules at prolonged emulsion duration, resulting in an insufficient amount of HDI to react with PEI to form a positively charged polyurea membrane. Therefore, the deposition of $\text{Si}(\text{OH})_4$ monomer and formation of tight silica shell are restricted.

The influence of reaction temperature on core fraction and thermal stability is shown in Fig. 3d. As can be seen, the core fraction of capsules reduces significantly from 81.9 to 43.1 wt.% from the surface of the microcapsules. The inset table in Fig. 4b reveals a high quantity of Si and O elements suggesting the successful deposition of silica as the shell material.

MTTS, deposition of $\text{Si}(\text{OH})_4$ monomers on the capsule surface are encouraged. At higher temperatures, condensation of $\text{Si}(\text{OH})_4$ monomers performs more quickly to generate silica shell, resulting in the content of silica shell increased in the capsules, *i.e.* core fraction of capsules under 45 $^{\circ}\text{C}$ about 43.1 wt.% is the lowest one among these three. The resultant capsules under 35 $^{\circ}\text{C}$ reaction temperature maintained at 80 $^{\circ}\text{C}$ for 1 h, the release of core octadecane was only about 3.4 wt.%. However, for capsules synthesized under reaction temperatures of 25 $^{\circ}\text{C}$ and 45 $^{\circ}\text{C}$, the release of core octadecane was 5.3 wt.% and 6.6 wt.%, respectively. Generally, a proper reaction temperature could promote the formation of a silica shell. Herein resultant capsules under a reaction temperature of 35 $^{\circ}\text{C}$ performed the best thermal stability. Under this circumstance, the hydrolysis product of MTTS is sufficiently absorbed on the capsule surface via electrostatic interaction. Gradually, a tight and stiff structure of silica capsule shell formed, which efficiently protected the leakage of core octadecane. However, the formation of shell structure appeared loosely and defectively under the unsuitable reaction temperature of 25 $^{\circ}\text{C}$ and 45 $^{\circ}\text{C}$, resulting in a higher release ratio of core octadecane under isothermal at 80 $^{\circ}\text{C}$ for 1 hour. It demonstrates that the hydrolysis and condensation of MTTS were more sensitive to the system temperature.

The influence of reaction time of 1 hour, 2 hours, and 3 hours on the resultant capsules was investigated and evaluated by a core fraction and thermal resistance of capsules. As can be seen in Fig. 3e, the core fraction of capsules reduced from 90.1 to 60.7 wt.% when the reaction time was prolonged from 1 hour to 3 hours. It can be explained that a large quantity of silica was deposited on the capsule surface with the reaction time increased from 1 hour to 3 hours. And thicker shell wall formed by silica results in the reduction of the core fraction. The core fraction of capsules decreased by about 10.3 wt.% after isothermal at 80 $^{\circ}\text{C}$ for 1 hour when the reaction duration was 1 hour. When the reaction time increased to 2 hours, the release of core octadecane was about 4.2 wt.% at the same condition. Subsequently, the reaction time was extended to 3 hours, and the core fraction of capsules decreased only about 3.4 wt.% after isothermal at 80 $^{\circ}\text{C}$ for 1 hour. This good thermal resistance phenomenon was ascribed to the denser shell wall structure of capsules to decrease the evaporation of core PCM. Therefore, the optimal capsules can be obtained when the reaction duration was 3 hours.

3.3 Morphology and physical properties of SiO_2 -PCM microcapsules

A digital image of SiO_2 -PCM microcapsules is shown in Fig. 4a, which appears like white powder. And it is made of SiO_2 . As can be seen in Fig. 4b, it presents the EDS spectra probed from the surface of the microcapsules. The inset table in Fig. 4b reveals a high quantity of Si and O elements suggesting the successful deposition of silica as the shell material.

In order to observe the microstructure of capsules clearly

Fig. 4 Digital photograph of (a) SiO_2 -PCM capsules and (b) EDS of SiO_2 -PCM capsules; and SEM images of SiO_2 -PCM microcapsules: (c) overview of SiO_2 -PCM capsules, (d) individual microcapsule, (e) shell profile, (f) overview of sintered SiO_2 -PCM capsules, (g) sintered individual microcapsule, and (h) sintered shell profile.

SEM was introduced as can be seen in Fig. 4c. The capsules are in spherical shapes with a diameter of about $200 \pm 15 \mu\text{m}$ (estimated using Software ImageJ to count at least 100 sample capsules). The surface of the capsule presents coarsely which may be ascribed to the random deposition of silica precursor and thus forming a rough surface with protruding nubs-like morphology (Fig. 4d). Well-defined core-shell structure with a very dense shell of about $10 \mu\text{m}$ in thickness without any defects can be observed in Fig. 4e.

To check the strength and stability of the silica shell, Fig. 4f shows the morphology of sintered SiO_2 -PCM microcapsules. It reveals that most of the heat-treated microcapsules are still in a spherical shape. However, few micro-holes appear on the shell due to evaporation of PCM during the sintering process (Fig. 4g). Shell thickness of sintered capsules remains unchanged at approximate $10 \mu\text{m}$ (Fig. 4h) which indicates that the pristine capsule shell almost consists of inorganic silica.

In order to verify that PCM is successfully microencapsulated, the core and shell materials are only physically bonded without chemical reaction, FTIR spectra were used to characterize different bonds of shell material, octadecane (core material), and SiO_2 -PCM capsules are shown

Fig. 5 FTIR spectra of shell, SiO_2 -PCM capsules, and octadecane.

in Fig. 5. For the shell, peaks at 778 and 1050 cm^{-1} are ascribed to the bending vibration of Si-O-Si.^[28] The peak at around 3400 cm^{-1} belongs to stretching vibration and bending the vibration of the functional group of OH.^[23] Peaks at 1584

Fig. 6 (a) TGA curves of shell, SiO₂-PCM capsules, and octadecane at a heating rate of 10 °C/min in N₂ environment; and (b) isothermal performance of pure SiO₂-PCM capsules and octadecane heated at 80 °C for 1h.

and 1626 cm⁻¹ can be assigned to the bending vibration of NH₂, which is indicative of the existence of polyurea monomer.^[14] Moreover, peaks at 1376, 2850, and 2915 cm⁻¹ correspond to the stretching vibration of C-H(CH₃), the peak at 1458 cm⁻¹ belongs to the deformation vibration of C-C, and the peak at 718 cm⁻¹ is attributed to the plane rocking vibration of CH₂.^[23,29] For the core octadecane, three typical intensive peaks at 1465, 2850, and 2930 cm⁻¹ are associated with the alkyl C-H stretching vibrations of methylene and methyl groups.^[21,30] The FTIR spectrum of SiO₂-PCM capsules seems to be the superposition of the shell and the core octadecane spectra. No new peak was observed in the spectrum of SiO₂-PCM capsules which indicates no chemical interaction between the core material and the shell during the synthesis of SiO₂-PCM capsules.^[31-33]

The above FTIR characterization can only indicate the success of microencapsulation, but it cannot confirm the integrity of the PCM microcapsule and the compactness of the shell material. Therefore, TGA is needed to further test its performance. TGA is mainly used to characterize the thermal decomposition and then judge the quality of the shell material of the microcapsule. Fig. 6a shows the TGA of octadecane, silica shell, and SiO₂-PCM capsules. As can be seen, octadecane starts to decompose at about 130 °C and is fully decomposed at 230 °C. Two degradation stages can be identified in the TGA curve of the silica shell, which is attributed to further dehydration and oxidation of the methyl group at higher temperatures leading to the removal of the OH group. For SiO₂-PCM microcapsules, three decomposition steps were observed in the TGA curve. A small weight loss of 5 wt.% before 130 °C is associated with the evaporation of residual water absorbed on the silica surface. Subsequently, a drastically weight loss between 130 °C and 230 °C is attributed to the evaporation and decomposition of the core material. Another degradation stage is after the decomposition of octadecane and up to 250 °C which is due to the decomposition of the polyurea membrane. At a much higher temperature

range of 300 °C -350 °C, a weight loss of about 7 wt.% is attributed to further dehydration from the condensation of silica shell at the higher temperature. It reveals that the condensation of silanol is not thorough and there is still some silanol with OH groups adsorbed on the surface of microcapsules.^[34] Fig. 6b shows the isothermal performance of octadecane and SiO₂-PCM capsules. As can be seen, octadecane loses approximately 31 wt.% of its weight through evaporation at 80 °C for 1h while the SiO₂-PCM capsules can maintain the weight without significant weight loss (reducing about 3.4 wt.%). The silica shell therefore effectively prevents leakage of the core material.

The thermal stability of capsules can also be tested by DSC in addition to TGA. The principle is that the integrity and compactness of the microcapsule shell material can be proved from the side if there is no change in the enthalpy value before and after several DSC cold and hot cycles. Fig. 7a shows the DSC curves of octadecane, PUA-PCM capsules, and SiO₂-PCM capsules. It obviously indicates that the PUA-PCM capsule's curve shifts seriously compared with that of octadecane, meaning that PUA-PCM capsules existed hysteresis phenomenon when PCM melting and solidifying occurred. This phenomenon corresponded to the lower thermal conductivity property of the polymer shell. In contrast, the curve of silica-PCM capsules is relatively higher and narrower as can be seen in Fig. 7a, which indicates silica has a relatively higher thermal conductivity. It clearly reveals that the curve of SiO₂-PCM capsules narrows than that of PUA-PCM capsules, indicating the high-temperature sensitivity of silica shell-based capsules. In addition, both the initial point of melting and solidifying of silica-PCM capsules were prior to those of PUA-PCM capsules octadecane due to the higher thermal conductivity of the silica shell. It is universally known that silica thermal conductivity, about 1.296 Wm⁻¹k⁻¹, is much higher than that of polymer (0.4 Wm⁻¹k⁻¹).^[35] Therefore, on the basis of inorganic silica shell, energy can transfer rapidly between capsules and ambient surroundings during the heating

Fig. 7 (a) DSC process of octadecane, PUA-PCM capsules, and SiO₂-PCM capsules; and (b) DSC melting-solidification cycles of SiO₂-PCM capsules at a heating rate of 10 °C/min in N₂ environment.

or cooling process, resulting in a narrow and high DSC curve.

DSC curves of SiO₂-PCM microcapsules remain largely unchanged even after 150 heating and cooling cycles (Fig. 7b) suggesting the excellent durability and reliability of the capsules. The slight shift of the peak melting and solidification temperatures was observed after a few heating and cooling cycles, which may be attributed to supercooling during phase change. It can be explained as follows. At the first heating-cooling cycle, PCM in microcapsules was solid at the beginning of heating, and the portion of PCM that touches the internal wall of microcapsules is solid as well. After the solidification process, the core PCM phase changed to solid again. Nevertheless, the same portion of PCM contacting internal wall of microcapsules still exists as liquid partially. This phenomenon appears from the second heating-cooling cycle. It reveals that the latter cycle needs relative less heat energy than the former cycle to make core PCM to change

from solid to liquid. This specific structure also makes microcapsules response more rapidly and sensitively for heat transfer.

3.4 Mechanical strength of SiO₂-PCM microcapsules

Figure 8a illustrates a typical compressive load-displacement curve of a single SiO₂-PCM capsule with a diameter of $300 \pm 20 \mu\text{m}$ (Fig. 6a inset). As can be seen, the compressive load increases gradually in a bi-linear manner. After the peak load, the capsule fractures and softens with a descending of the load carrying capacity. Fig. 8b compares the mechanical strength of a single SiO₂-PCM microcapsule and a single PUA-PCM capsule of different sizes. As can be seen, the strength of both capsules reduces with increasing size. A similar trend has been reported for other polymer shell capsules.^[36] Moreover, the mechanical strength of SiO₂-PCM capsules under compression is much higher than that of PUA-PCM capsules

Fig. 8 (a) Typical compressive load vs. displacement curve of individual SiO₂-PCM microcapsule with a diameter of 300 μm; and (b) compressive strength of microcapsules as a function of capsule size.

(about 3.5 times in the current study). Compared to published literature, the strength of 100 μm SiO_2 -PCM capsules in the current study reaches 27.6 MPa, which is about 6 times that of a polyurethane microcapsule (4.7 MPa) with the same diameter.^[24] and higher than that of a hollow glass bubble (12.5 ± 2.0 MPa) with a smaller diameter of around 70 μm .^[37] The dense inorganic silica shell greatly enhances the mechanical strength of SiO_2 -PCM microcapsules.

4. Conclusion

SiO_2 -PCM microcapsules were fabricated successfully via interfacial polymerization and electrostatic interaction. The resultant microcapsules were characterized through the SEM, TGA DSC, FTIR, and compressive strength methods. SEM images show that compact and intact silica shell-based PCM capsules were obtained, and the collected capsules were free-flowing. TGA curves indicate that the final residual weight percentage of capsules was about 20 wt.% after sintered at 600 °C in a nitrogen atmosphere. A high encapsulation ratio, corresponding to energy storage capacity, about 90 wt.% with a capsule diameter of 500 μm , was calculated by enthalpy or alternative physical experiment method. Long term performance of silica-PCM capsules investigated via 150 heating-cooling cycles, showed either the initial (1st cycle) or final (150th cycle) enthalpy of silica-PCM capsules nearly had no changes was ascribed to compact and intact silica shell to inhibit core PCM evaporation and spillage out of capsules. Silica-PCM capsules performed better temperature sensitivity than PUA-PCM capsules due to the high thermal conductivity of silica shells, resulting in a narrow melting and solidification range and high energy storage efficiency. Compression testing indicated that the single silica-PCM capsule is robust with a maximum compressive strength (27.6 MPa, size: 100 μm) and much higher than that of a polymer shell-based capsule. In conclusion, a facile method to fabricate robust and stable silica-PCM capsules had been developed, silica-PCM capsules probably have the potential commercial applications in energy storage and energy saving.^[38-41]

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Supporting Information

Not Applicable.

Conflict of Interest

There is no conflict of interest.

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